

Xiu en Zhang and Ale Zunger

(Received 3 March 2010; published 15 June 2010)

We show how pressure fundamentally alters the repulsive nature of the nonreactive Li-Cs mixture, converting it from strong phase separating at ambient pressure to strong long-range ordering at high pressures. The ordered phases found via a global space group optimization within the density-functional theory are  $\text{Li}_7\text{Cs}$  in the  $Cmmm$  structure,  $\text{LiCs}$  in the  $B2$  structure, and  $\text{Li}_7\text{Cs}$  in the  $C2Li5$  structure. These structures are remarkably stabilized by a pressure-induced increase in charge transfer from Cs to Li unit, an unusual effect concerning two elements from the same group (isovalent). These high-pressure phases exhibit interesting behaviors: (i)  $\text{LiCs}$  ( $B2$ ) has its Cs  $5p$  core states nearly degenerate with the valence states at ambient pressure [11–13] and our density-functional-calculated formation enthalpies ( $\Delta H$ 's) summarized in Fig. 1 show that Na

$\text{Li}_2\text{Cs}$ ,  $\text{Na}_2\text{K}$ , and  $\text{K}_2\text{Cs}$  form ordered Laves phase ( $\text{MgZn}_2$  type). Yet, the Li interalkalies clearly exhibit only phase-separation behavior with concomitant calculated formation enthalpies that are positive and rapidly increase with the size mismatch, i.e., 56, 179, 215, and 227 meV/atom for Li-Na, Li-K, Li-Rb, and Li-Cs, respectively. Li-Cs with its largest positive  $\Delta H$  is the paradigm phase-separation interalkali due to its large atomic size mismatch of  $\Delta V/\bar{V} \approx 142\%$ . Pressure can change non-reactivity tendencies [17] as illustrated by Li-Be (high, however, is a nonisovalent combination more prone to reactivity 3(Pressure 79.4(see Fig. 6) in Cs) Tj/F21 Tf1.53920TD(0)Tj/F51 Tf0.33060TD(C)Tj/F41 Tf0.68830TD(2)Tj/F51 Tf0.4986

Li, the *hR1* structure transforms to the *cI16* structure at about 35 GPa, then at about 75 GPa it transforms to the *C2* structure, and then at about 98 GPa it transforms to the *Aba2* structure, which persists up to at least 160 GPa; for Cs, the *Cmca* structure transforms to the *dhcp* structure at about 60 GPa, which persists up to at least 200 GPa. The transition pressures are in agreement with the experimental results [3–5]. Random search techniques [24] and the evolutionary algorithm method [25,26] were previously applied to search the ground states of elemental lithium [6,7] and cesium [8,9], and cesium [7], respectively, and also Ceban [3,4] (experimental).

sl:78.653(e18824 0.(,l)-421)D7.65

$$\Delta H_{\text{VD}} = \frac{1}{4} (H_{\text{LiCs}} - H_{\text{Li}} - \frac{1}{2} H_{\text{Cs}}) + P(V_2 - V_1)/2 \quad (1)$$

The charge-transfer energy  $\Delta H_{\text{CT}}$  is the enthalpy change when combining bcc Cs and bcc Li at the constant-volume  $V_m$  to form LiCs (B2),

$$\Delta H_{\text{CT}} = H_{\text{LiCs}} - H_{\text{Li}} - H_{\text{Cs}} \quad (2)$$

$\Delta H_{\text{VD}}$  and  $\Delta H_{\text{CT}}$  are shown as the red line and the green line in Fig. 4(a), respectively. We see that  $\Delta H_{\text{VD}}$  is positive and increases rapidly as pressure increases, the main effect coming from the  $P(V_2 - V_1)/2$  term (dashed line). At zero pressure,  $\Delta H_{\text{VD}}$  (520 meV/atom) is large enough to make the Li-Cs system phase separating. In contrast,  $\Delta H_{\text{CT}}$  is negative and decreases rapidly as pressure increases. At 160 GPa,  $\Delta H = -258$  meV/atom and  $\Delta H_{\text{CT}}$  is large enough to induce ordering in Li-Cs.

The electronic structures of LiCs (B2) are shown in Figs. 4(b) and 4(c). At zero pressure the dispersions of the  $\text{Li } d_{1s}$ ,  $\text{Cs } d_{5s}$ , and  $\text{Cs } d_{5p}$  bands are all small. At 160 GPa, the  $\text{Li } d_{1s}$  band still has negligible dispersion, while the dispersion of the  $\text{Cs } d_{5s}$

high-pressure compounds between two elements in the same column of the periodic table (Li-Cs) is perhaps reminiscent of compound formation and charge transfer between Ag-K [27] or Li-Be [17], although the latter two cases are perhaps less surprising than Li-Cs as they represent a far larger chemical disparity (hence, reactivity) between the constituent elements that belong to different columns in the periodic table.

$\text{Li}_7\text{Cs}$  ( $Cmmm$ )<sup>x</sup>. . . 1D- . . . The electronic structure of  $\text{Li}_7\text{Cs}$  ( $Cmmm$ ) at 80 GPa exhibits interesting features of 1D-like bands. Its density of states at the Cs  $5s$  and Cs  $5p$  bands [see Fig. 6(a)] displays a remarkable feature of typical 1D electronic structure, the emergence of which in a three-dimensional environment is rather unexpected. Figure 6(b) shows the band structure of  $\text{Li}_7\text{Cs}$  ( $Cmmm$ ) at 80 GPa. The Cs  $5s$  and Cs  $5p$  bands display strong dispersion only along the  $\Gamma$ - $X_3$  direction, but are nearly dispersionless along other directions. The 1D band structure is caused by the unique crystal structure of  $\text{Li}_7\text{Cs}$  ( $Cmmm$ ) [see the inset of Fig. 3(a)]. The Cs chains are embedded in the three-dimensional Li networks and are separated from each other. Along the chain direction, the Cs atoms are close to each other and their cores overlap.