

Metal-Dimer Atomic Reconstruction Leading to Deep Donor States of the Anion Vacancy in II-VI and Chalcopyrite Semiconductors

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First-principles *ab-initio* calculations reveal a novel local atomic reconstruction mode of anion vacancy in II-VI and chalcopyrite compounds, leading to the formation of metal dimer. As a consequence, the neutral Se vacancy has a new electronic state in ZnSe and $6.6410.5257 \text{ C5}(\text{of})\text{-}2\text{Zn2}(\text{con})\text{m17eac}$ ic

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Vacancies are the most fundamental point defects in solids, controlling mechanical properties in metals [1]

to determine for the anion vacancy the atomic configuration, the electronic configuration (single-atomic defect or pair), defect formation energy H

and optical transition energies. The *ab-initio* E and atomic forces were calculated in the *ab-initio* total-momentum space formalism [16]. Within the local density approximation (LDA) of density functional theory, we use the *ab-initio* LDA exchange correlation potential as implemented by Perdew and Zunger [17] and projected onto a plane-wave basis set implemented in the VASP code [18]. The energy cutoff in the plane-wave expansion is 280 eV, and Brillouin zone integration is performed on a $3 \times 3 \times 3$ mesh, using the method of Monkhorst-Pack. In case of the charged vacancy, the electrostatic interaction of periodic images is corrected by the method of Evjen [19], where L is the linear cell dimension. We use the empirical lattice constant of

ZnSe and CuGaSe₂ [20]. The LDA band gap is
corrected by acknowledging the cation d and

energy of the $a_1(a)$ orbital in ZnSe (CuGaSe₂) from $E_v + 1.1$ eV ($E_v + 0.2$ eV), before the a_1 ion, of $E_v + 0.2$ eV ($E_v - 2.5$ eV), after the a_1 ion. Thus, a similar trend is observed in the energy level of Fig. 1, after the a_1 ion, the double occupied a_1^2 level is located in ZnSe just above the VBM, whereas in CuGaSe₂ it is deep inside the valence band. When V_{Se}^0 is ionized, i.e., for the a_1^0 configuration of V_{Se}^{2+} (Fig. 1), having a break of the metal-metal dimer (Table I), and a change in the original T_d lattice symmetry (Fig. 1, bottom). The energy level of the a_1 state in ZnSe moves up from $E_v + 0.2$ eV (V_{Se}^0) to $E_v + 2.5$ eV (V_{Se}^{2+}), whereas in CuGaSe₂ it moves up from $E_v - 2.5$ eV (V_{Se}^0) to $E_v + 1.5$ eV (V_{Se}^{2+}) [20]. The atomic a_1 ion is the driving force for this level shift can be justified from the fact that the electrostatic effect, i.e., the relief of the electrostatic Coulomb repulsion of the $a_1^2(V_{Se}^0) \rightarrow a_1^0(V_{Se}^{2+})$ transition, would have displaced the a_1 level to a lower energy. Because of the charge dependence of the defect level from below VBM to above the VBM, the Se vacancy in chalco-selenides can accommodate a conduction electron, in which electron configuration from the VBM to the deep defect level, i.e., filling the hole. Thus, the Se vacancy can explain the energy ho-

