Is there an elastic anomaly for a (001) monolayer of InAs embedded in GaAs?

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Di V`]g\YX'Vm'h\Y'5=D'Di V`]g\]b[

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Is there an elastic anomaly for a (001) monolayer of InAs embedded in GaAs?

James E. Bernard

National Renewable Energy Laboratory, Golden, Colorado 80401

When a coherently grown (001)-oriented layer of InAs is embedded in a GaAs host, the coherency

theory to be $\epsilon_{\perp}=7.3\%$. Brandt, Ploog, Bierwolf, and Hohenstein, [Phys. Rev. Lett. **68**, 1339 (1992)]

report on an investigation into whether a first-principles local-density total energy minimization shows such an elastic anomaly in the monolayer limit. We find that it does not.

When a layer of material with cubic lattice parameter a_0 is grown coherently in a host with cubic lattice parameter $a_h \neq a_0$, the buried layer undergoes a distortion of its lattice parameter c perpendicular to the plane of the layer in response to the distortion of its lattice parameter parallel to the

bulk energy as a function of all the structural degrees of

harmonic continuum elasticity theory (see, e.g., Ref. 1). For

to (001), the perpendicular strain $\epsilon_{\perp}=(c-a_0)/a_0$ is related[1] to

$$\epsilon_{\perp}=-2\frac{C_{12}}{C_{11}}\epsilon_{\parallel}, \quad (1)$$

For example, with the measured[2] $C_{11}=0.8329$ and $C_{12}=0.4526$ (Tdyn/cm^2) for InAs, Eq. (1) predicts $\epsilon_{\perp}=7.27\%$ for a layer of InAs buried in GaAs.

Recently, Brandt *et al.*[3] examined the strain of buried (001) layers of InAs in GaAs experimentally, via a high-

sponding to $\epsilon_{\perp}=12.46\%$, much greater than the prediction of

plied to a *three* monolayer film of InAs revealed a lattice

ness of the embedded layer. Regarding (1), we can expect the

InAs ($\approx 7\%$) is not sufficient to cause a substantial departure from harmonicity. Regarding (2), it was previously shown[4] that for certain interface orientations nonuniform atomic relaxations near an interface can be large. But for (001)-oriented interfaces the allowed relaxations all lie parallel to [001], and the nonuniformity of these interplanar relaxations

picture provided by continuum elasticity theory. Regarding

been untested in semiconductor layers.[5] One might imagine,

small, interaction between them might substantially alter the

theory.

In view of the unprecedented result of Brandt *et al.*,[3] we tested the validity of Eq. (1) via an atomistic, nonharmonic, first-principles theory of the relaxations of such a system to

demonstrated in the experiment. Our first-principles results in no way depend upon Eq. (1) or the approximations on which it is based. We found good agreement with Eq. (1) and no theoretical evidence for such a breakdown, in spite of

Our tests improve on harmonic continuum elasticity

proach by a discrete atomistic approach, still limited to

tions to the total energy, not just the elastic contribution

mation (LDA). With the total energy represented as a functional of the electron density of the system, all contributions

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to it are treated on the same quantum-mechanical footing. This avoids all the potential pitfalls discussed above, and can be regarded as giving the best available theoretical estimate

In the first step, we applied the VFF model, with the

bond lengths, to the calculation of the relaxed geometry of a structure containing an embedded (001)-oriented layer of InAs in a GaAs host, all confined to the GaAs lattice constant in the (001) plane. We found values ranging from $\epsilon_{\perp}=7.09\%$ with 1 monolayer (ML) of InAs, to $\epsilon_{\perp}=7.17\%$ with 10 ML.[9] This demonstrates good agreement with the prediction of Eq. (1) and shows no evidence of anomalously large distortion in the ultrathin limit.

We note that in a recent paper Massies and Grandjean[10] used a one-dimensional variant of the VFF model to investigate the behavior of monolayer-height InAs surface islands on GaAs, finding a relaxation away from the surface larger

ever, the clear agreement of our bulk three-dimensional VFF calculations with continuum elasticity theory demonstrates that this comparison of a one-dimensional model of surface relaxation with bulk strain is inappropriate.

The inability of the VFF model to reproduce the

include electronic effects as well. Our total-energy calculations were performed using the LDA in the form of the

equivalent of 10 special points[14] in the irreducible zinc-

5%, far less than the more than 70% difference between the

stants of zinc-blende GaAs and InAs were determined by fitting to a Birch equation of state[15] the total energies from seven calculations with differing lattice constants. The fit gives the values (Å) 5.60±0.005 (5.65) for GaAs and 6.02 ±0.005 (6.06) for InAs, where experimental values[2] are

constants induces only about a 4% (0.3 percentage points) overestimation in our calculated $\epsilon_{\perp}$.

For the first-principles calculations of the properties of

to obtain the structural effect of a thick GaAs substrate. Tests

more than adequate to converge the structural features of the interface and the GaAs layer to match those with much thicker layers. We minimized the SL total energy with respect to all structural degrees of freedom, subject to the coherency constraint. Quantum-mechanical forces[11] were used to relax the atomic positions, and a fit of total energy vs c/a_h

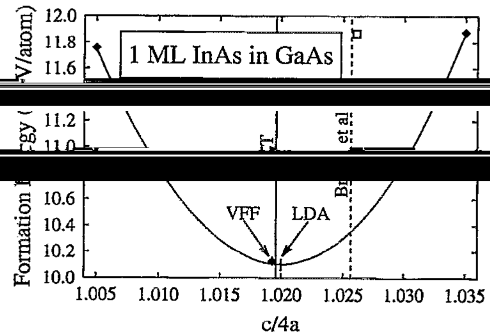

FIG. 1. Formation energy of the atomically relaxed eight-layer structure containing 1 monolayer of InAs vs its $c/4a$ ratio. Filled diamonds denote direct first-principles LDA results; the curve is a cubic fit to those points. The minimum of the fit curve is marked by a vertical tick, labeled "LDA." The diamond symbol labeled "VFF" lies at the $c/4a$ value predicted by the VFF model. The solid vertical line, labeled "CET," marks the prediction of

the InAs layer has $\epsilon_{\perp}=12.5\%$, illustrating the large energy penalty incurred if the strain is concentrated in the InAs layer.

was used to find the equilibrium c/a_h ratio. While the symmetry of the structure permits additional relaxations beyond

these was found to be rather small.

Our results are illustrated for the single-monolayer case

lattice constants). We estimate the error in the LDA value

recent pseudopotential calculation by Shiraishi and Yamaguchi.[16] However, correspondence with one of the authors[17] has clarified that their results do not, in fact, show evidence of an elastic anomaly in the monolayer limit.

so because the determination of $\epsilon_{\perp}$ is based on the overall measured shift in the lattice.[9] Because the VFF model provides a sufficiently accurate reproduction of the structural

respectively, can raise $\epsilon_{\perp}$ to 13.05%, comparable to the

refinement of the analysis of the HREM data of Ref. 3, permitting more precise determination of the individual spacings, suggests that they should be able to detect excess In, at least in the two larger amounts we have suggested.[20]

first subsequently grown GaAs layer would be to change the

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ratio of the two expanded spot-row spacings in different

[110] or along [110]—in one case increasing the ratio, and in the other case decreasing it. Thus it would be possible to compare such images as another test for excess In. To our

of InAs to be distributed among three layers, with 75% in the

nately, the samples used for this study were not characterized by HREM, so it is not clear whether they would exhibit the

principles total-energy methods to the determination of the

ment to help determine whether the experimental results of Brandt *et al.*[3] could be a consequence of excess In in the

mental findings, and further work to attempt to resolve the

One of us (J.E.B.) acknowledges support for this work received under Subcontract No. XR-1-11236-1 from the National Renewable Energy Laboratory. Work at NREL

gratefully acknowledged. We thank the reviewer for pointing

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generally at bulk semiconductor interfaces (Ref. 4).

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InAs in GaAs (2.55 Å). However, Dr. Shiraishi has informed us that the

blende GaAs bond length. Their calculated bond length for the embedded layer is 2.49 Å, corresponding to $\epsilon_\perp=4.6\%$.

with the overall shift. Further, application of the method (Ref. 19) to

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